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### **A First Principle Study of Structural and Electronic Properties of Bulk MoS<sub>2</sub> and Its Monolayer**

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#### **ABSTRACT**

In this study, we have investigated the structural and electronic properties of bulk and monolayer MoS<sub>2</sub> using a first principle method based on density functional theory. The indirect band gap in the bulk MoS<sub>2</sub> was found to be 0.975 eV, whereas in the monolayer MoS<sub>2</sub> the band gap of 1.85 eV was found to be direct one. The calculated physical parameters of monolayer MoS<sub>2</sub> are found to be very close to the bulk MoS<sub>2</sub> and compare well with available experimental and theoretical results. The calculated density of states (DOS) may help explain this change in the nature of band gap in bulk and in monolayer MoS<sub>2</sub>.

**KEYWORDS :** MoS<sub>2</sub>, Electronic properties, monolayer, Band Structure, Density of States.

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## 1. INTRODUCTION

Transition metal dichalcogenides (TMDs) possess layered structures<sup>1</sup>. Layered transition-metal dichalcogenides (LTMDs) have been extensively reviewed in the recent past<sup>2-4</sup>. MoS<sub>2</sub> is a typical example of LTMD family of materials which attracts investigation because of its distinctive industrial applications, ranging from use as a lubricant<sup>5-6</sup> and a catalyst<sup>7</sup> as well as in photo-voltaics<sup>8</sup> and energy storage<sup>9</sup>. In its bulk form MoS<sub>2</sub> is a semiconductor with an indirect band gap of about 1.23 eV while its monolayer has a direct energy gap of 1.8 eV<sup>10</sup>. A special attention has been paid on single layer MoS<sub>2</sub> in the recent years. Upon thinning from the bulk the electronic structure of MoS<sub>2</sub> undergoes an interesting transition<sup>11-13</sup>. Recently, a monolayer MoS<sub>2</sub>-based field effect transistor (FET) with HfO<sub>2</sub> as gate insulator has been successfully implemented<sup>14</sup>. These ideal properties make monolayer MoS<sub>2</sub> a very promising candidate for next generation FET and as optoelectronic devices<sup>15</sup>. This has raised enormous interest in exploring the extraordinary properties of mono layers of MoS<sub>2</sub>.

Theoretically, there are various possibilities of energy gap manipulation in MoS<sub>2</sub>. By reducing the layer thickness from bulk to monolayer, the indirect band gap energies in the bulk are shifted relative to the direct band gap in the monolayer limit<sup>16</sup>. It undergoes a transition from an indirect to direct gap exhibiting strong photoluminescence when confined in a 2D monolayer<sup>17</sup>. It has been suggested a way to engineer 3D semi-conducting MoS<sub>2</sub> nano particles with direct band gaps as well as metallic dichalcogenides nanowires with promising catalytic and thermoelectric properties<sup>18</sup>. Eellis *et al.* carried out a study using HSE screened hybrid functional and offered improvement over semi local density functional. All electron calculations including spin orbit coupling were performed and confirmed indirect to direct band gap transition<sup>19</sup>. Electronic structure of transition metal dichalcogenides has been studied using *ab initio* theory using Troullier-Martin norm conserving, relativistic pseudopotentials in fully separable Kleinman and Bylander form<sup>20</sup>. They used exchange and correlation energies within LDA.

The properties of transition metal dichalcogenides (TMDs) not only can be tuned by varying number of layers, but also can be modified by application of external field or strain engineering. Studies<sup>21-22</sup> have confirmed that applying strain is one of the best possible strategies to tune the band gap, since it neither attenuates the properties nor is inefficient for single layers. It has been predicted that straining MoS<sub>2</sub> modifies the band gap energy and the carrier effective mass. Moreover, at strains larger than 1% the lowest lying band gap changes from direct to indirect<sup>23-26</sup>. It has been suggested that strain engineering of the band structure of MoS<sub>2</sub> could be used to increase carrier mobility of MoS<sub>2</sub>, to create tunable photonic devices and solar cells<sup>27</sup> and even to control the

magnetic properties of MoS<sub>2</sub>. While strain perturbs the band structure of all materials, two-dimensional materials such as MoS<sub>2</sub> can sustain strains greater than 11% allowing exceptional control of material properties by strain engineering<sup>28</sup>. In a recent study the conduction band valley structure of a few layer MX<sub>2</sub> by close examination of temperature dependent indirect excitation emission peaks has been explored<sup>29</sup>. A study on elastic constants and electronic structures of two-dimensional monolayer MoS<sub>2</sub> under elastic strain using first principle calculations has been made<sup>30</sup>. It is shown that the band gap of monolayer MoS<sub>2</sub> undergoes a descent trend with the increase in strain. They observed a direct to indirect transition at strain of 0.01 and semiconductor to metal transition at strain of 0.10.

With the goal of understanding the electronic properties of bulk and monolayer MoS<sub>2</sub> and strain engineering, we carried out *ab initio* calculations of bulk and monolayer MoS<sub>2</sub> using gradient corrected exchange-correlation functional in DFT framework and observed a transition from indirect to direct band gap. However if this band gap can be tuned such that a semiconductor with a lower band gap or a semiconductor to metal transition can be achieved with the application of strain, then a wide range of tunable nano device can be fabricated. In the present work, therefore we study the effects of mechanical strains on the electronic properties of monolayer of MoS<sub>2</sub>. Our results suggest a way of band gap engineering in MoS<sub>2</sub>.

## 2. COMPUTATIONAL DETAILS

The calculations are performed by using density functional theory (DFT)<sup>31-32</sup> based Atomistix virtual nano lab (ATK-VNL) tool<sup>33</sup>. This method has been previously used to study the electronic properties of undoped and doped graphene<sup>34-35</sup>. The exchange correlation potential was approximated by generalized gradient approximation using Perdew-Wang 91 functional (GGA-PW91)<sup>36</sup>. The atomic positions and cell parameters were fully relaxed until an energy convergence of 10<sup>-9</sup> eV reached. We used wave function and charge-density cut-offs of 70 Ryd and 300 Ryd, respectively. First, we obtained lattice constants *a* and *c* by the process of total energy minimization. Optimized structure (coordinates) was used to perform self consistent calculations with a Monkhorst-Pack<sup>37</sup> 8×8×8 *k*-mesh followed by the non-self consistent calculations for band structures, density of states and partial density of states of bulk MoS<sub>2</sub>.

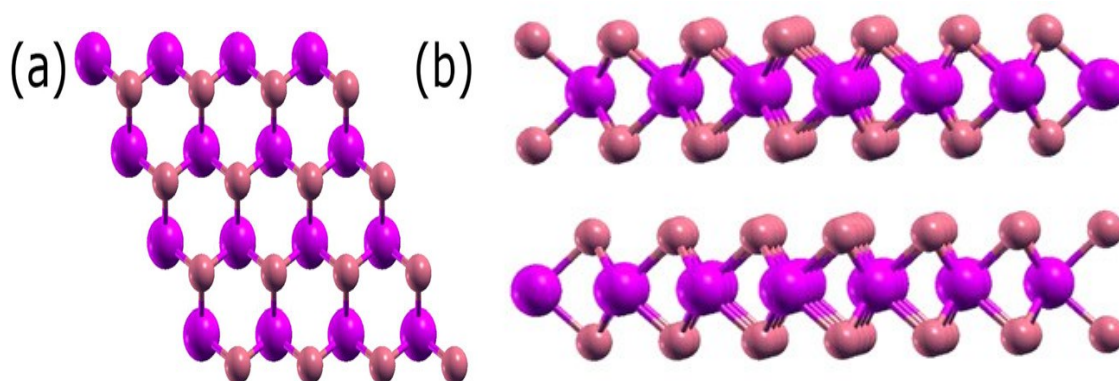
We used 80×80×80 *k*-points mesh along the path  $\Gamma$ -K-M- $\Gamma$  in the irriducible Brillouin zone to obtain the band structure with a very fine mesh points. However, for monolayer we used 8×8×1 and 80×80×1 Monkhorst-Pack of *k*-points respectively for sampling Brillouin zone for calculations of structural properties and electronic structure. In case of monolayer MoS<sub>2</sub>, we created 15 Å vacuum along *Z* axis to isolate it and to prevent any interaction between the layers. The

cohesive energy per atom of bulk MoS<sub>2</sub> was calculated as  $E_{\text{coh}} = E(\text{MoS}_2) - E(\text{Mo}) - 2E(\text{S})$ , where  $E(\text{MoS}_2)$  is the total energy of the unit cell of Molybdenum disulphide,  $E(\text{Mo})$  is the energy of Mo atom and  $E(\text{S})$  is the energy of S atom. The cohesive energies per atom of monolayer MoS<sub>2</sub> was also calculated accordingly. A uniform tensile strain ranging from 0 to 10% were applied on monolayer MoS<sub>2</sub> to study the change in behavior of its band gap.

### 3. RESULTS AND DISCUSSIONS

#### 3.1 Structural Properties

Molybdenum disulphide has a hexagonal structure consisting of S-Mo-S layers as shown in Figure 1.



**Figure 1.** Optimised geometric structures (a) Top view of bulk MoS<sub>2</sub>; (b) Side view of bulk monolayer MoS<sub>2</sub>. The Mo-atoms are denoted by purple and S-atoms by pink balls.

Bulk MoS<sub>2</sub> has two such layers and Mo atoms of one layer are directly above the sulfur atoms of the other layer and vice versa while monolayer MoS<sub>2</sub> has a single S-Mo-S layer. We have calculated the structural parameters of bulk and monolayer MoS<sub>2</sub> using GGA as shown in Table 1.

**Table 1.** Calculated structural parameters of bulk MoS<sub>2</sub> and monolayer MoS<sub>2</sub> using GGA. The available results have also been given for the purpose of comparison.

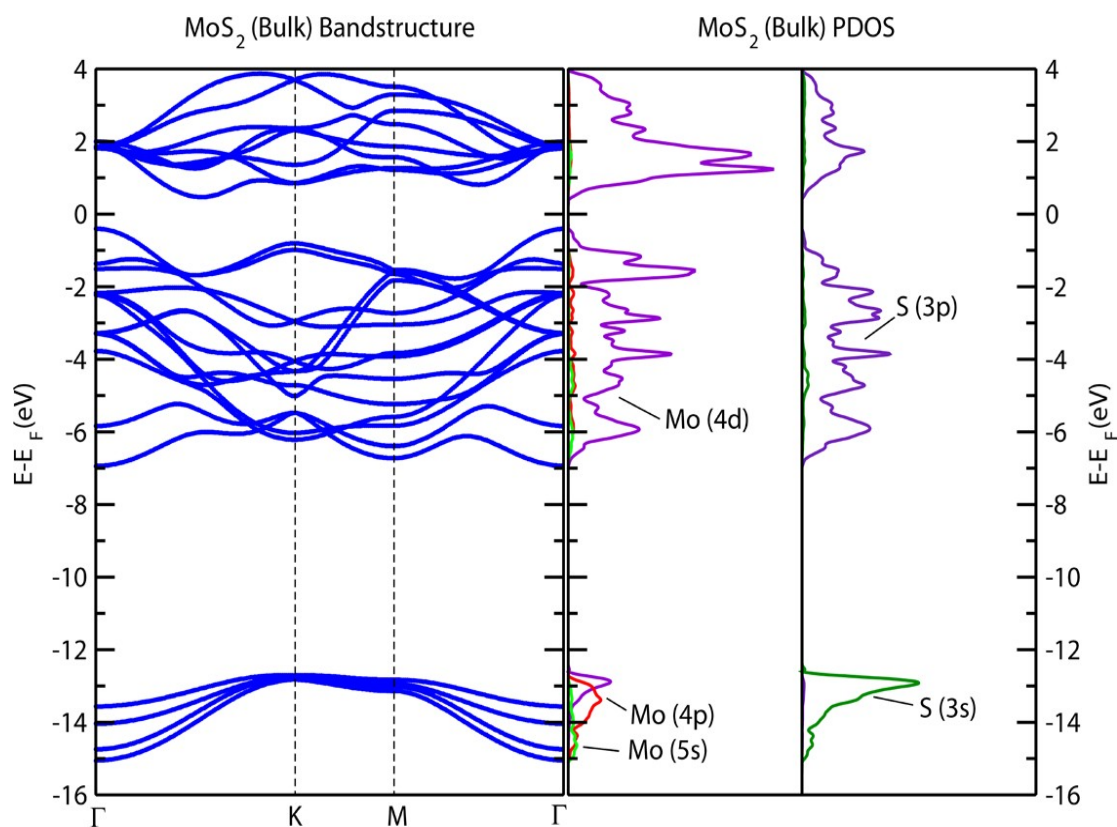
Properties		Bulk- MoS <sub>2</sub>	Monolayer- MoS <sub>2</sub>
Lattice Constant (Å)		3.185	3.185
Energy Gaps (eV)	Present calculation	0.975	1.850
	Experimental value	1.25 ( Ref 10 )	1.80 ( Ref 10 )
	Theoretical Value	0.89 ( Ref 20 )	1.55 ( Ref 20 )
Cohesive Energy (eV/atom)		4.955	4.969
Interatomic Distance d <sub>Mo-s</sub> (Å)		2.418	2.418

The results of bulk MoS<sub>2</sub> have been compared with experimental data while the results of monolayer are compared with some other theoretical results. We find excellent agreements as can be seen in Table 1. Our calculated lattice parameters overestimate the experimental values which is an

inherent feature of standard GGA functional. It is concluded that all the structural parameters calculated for monolayer MoS<sub>2</sub> are nearly identical to the structural parameters calculated for bulk MoS<sub>2</sub>

### 3.2 Electronic Properties

The electronic band structure and density of states can be divided into three sets of bands and states respectively, separated by gaps. In the first set, bands in electronic band structure and states in density of states around -14 eV are mainly due to 3s orbital of S atom separated by large gap from second set below Fermi energy in which 3p orbital of S and 4d orbital of Mo are mainly contributing and show strong hybridization. Third set above the Fermi energy in which main contribution is due to 4d orbital of Mo is separated by band gap from second group below Fermi energy. The electronic band structure of bulk MoS<sub>2</sub> and corresponding density of states are shown in Figure 2.

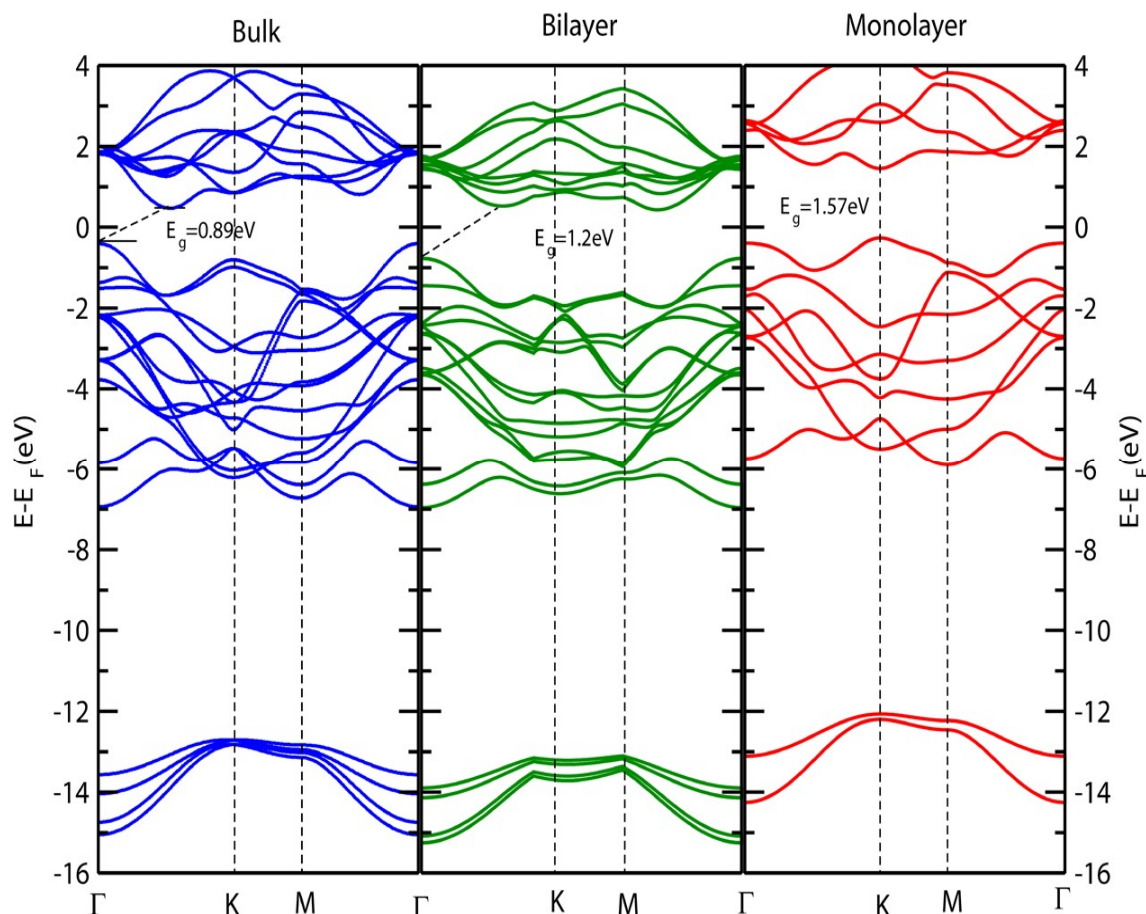


**Figure 2.** Calculated band structure (left panel) and orbital-projected density of states (PDOS) on Mo (middle panel) and S atoms (right panel) in bulk MoS<sub>2</sub>.

A comparative band structure of MoS<sub>2</sub> bulk and its monolayer and bilayer are shown in Figure 3. The bands on each side of the band gap are derived mainly from the 4d states of Mo and 3p states of S in bulk, bilayer and monolayer MoS<sub>2</sub>. The bands around the band gap are relatively flat, as expected from the *d*-character of electron states at these energies.

For bulk MoS<sub>2</sub> the valence band maximum is at high-symmetric  $\Gamma$ -point and conduction band

minima is in between  $\Gamma$ - and K-points, revealing indirect band semiconductor as can be seen in Figure 3. If we compare the band structure of bulk and monolayer MoS<sub>2</sub>, we observe that the band edge near  $\Gamma$  point has been shifted up by around 0.7 eV. In case of monolayer, at  $\Lambda$  and  $\Sigma$  point the band edge shifted up in such a way that the conduction band minima occurs at K-point. For monolayer the valence band maxima and conduction band minima are both at high-symmetric K-point revealing direct band gap semiconductor as can be seen in Figure 3.



**Figure 3.** Calculated band structure of bulk (left), bilayer (middle) and monolayer (right) of MoS<sub>2</sub> at high-symmetric points in the irreducible Brillouin zone. The position of valence band maxima, conduction band minima and the band gap ( $E_g$ ) are indicated. For monolayer MoS<sub>2</sub> the direct band gap occurs at K-point, unlike in other cases.

Thus there is a transition from indirect band gap to direct band gap as we go from bulk MoS<sub>2</sub> to its monolayer. A PDOS comparison of bulk and monolayer MoS<sub>2</sub> (as shown in the Figure 4) reflects that the states are essentially due to  $d_{z^2}$  and degenerate states  $d_{xy}$  and  $d_{x^2-y^2}$ . The states  $d_{x^2-y^2}$  and  $d_{xy}$  are degenerate in case of bulk while little bit separating in monolayer. The calculated and measured band gap for monolayer MoS<sub>2</sub> and bulk. Our calculated band gaps are in good agreements with other theoretical values.

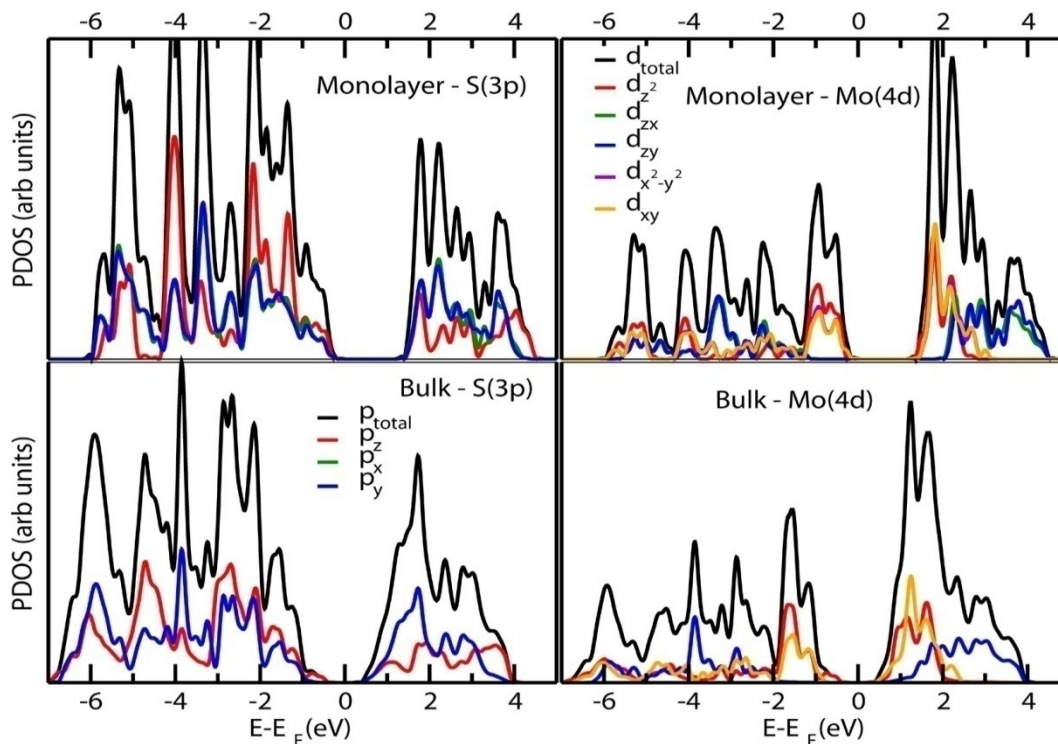


Figure 4. Calculated projected density of states (PDOS) of bulk and monolayer of MoS<sub>2</sub> are shown for Mo(4d) and S(3p) states. The legends for the *p*- and *d*-orbitals are similar for the monolayer and bulk.

#### 4. CONCLUSION

In summary, we have studied the structural and electronic properties of MoS<sub>2</sub> using plane wave pseudopotential method under GGA scheme based DFT calculations. Electronic band structure and density of states calculation show many similarities between monolayer-MoS<sub>2</sub> and bulk-MoS<sub>2</sub> except the nature of the band gap which is found direct for monolayer-MoS<sub>2</sub> as compared to indirect for bulk-MoS<sub>2</sub>. This observation is consistent with the theoretical prediction of indirect to direct band gap transition in going from bulk to monolayer. Such behavior, arising from *d*-orbital related interaction in MoS<sub>2</sub>, may also arise in other layered transition metal dichalcogenides. A further variation in band gap has been observed in MoS<sub>2</sub> monolayer on applying strain. It points out a new direction of band engineering hence such capability can lead to engineering novel behaviors and holds promise for new applications.

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